

Fall 10-4-2015

# Orientation dependence of dislocation transmission through twin-boundaries studied by in situ $\mu$ Laue diffraction

Nataliya Malyar

*Max-Planck-Institut*, malyar@mpie.de

Nagami Jaya

*Max-Planck-Institut*

Gerhard Dehm

*Max-Planck-Institut*

Christoph Kirchlechner

*Max-Planck-Institute on Research of Collective Goods*

Jean Sebastuen Micha

*Grenoble University*

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## Recommended Citation

Clarck W.A.T., Wagoner, R.H., Shen, Z.Y. (1992): On the criteria for slip transmission across interfaces in polycrystals – In: Scripta Metallurgica et Materialia, 26, 203-206. Imrich P.J, Motz C., Dehm G., Kirchlechner C. (2014): Differences in deformation behavior of bicrystalline Cu micropillars containing a twin boundary or a large-angle grain boundary – In: Acta Materialia, 73, 240-250. Jin Z.-H., Gumbsch P., Ma E., Albe K., Lu K., Hahn H., Gleiter H. (2006): The interaction mechanism of screw dislocations with coherent twin boundaries in different face-centered cubic metals – In: Scripta Materialia, 54, 1163-1168.

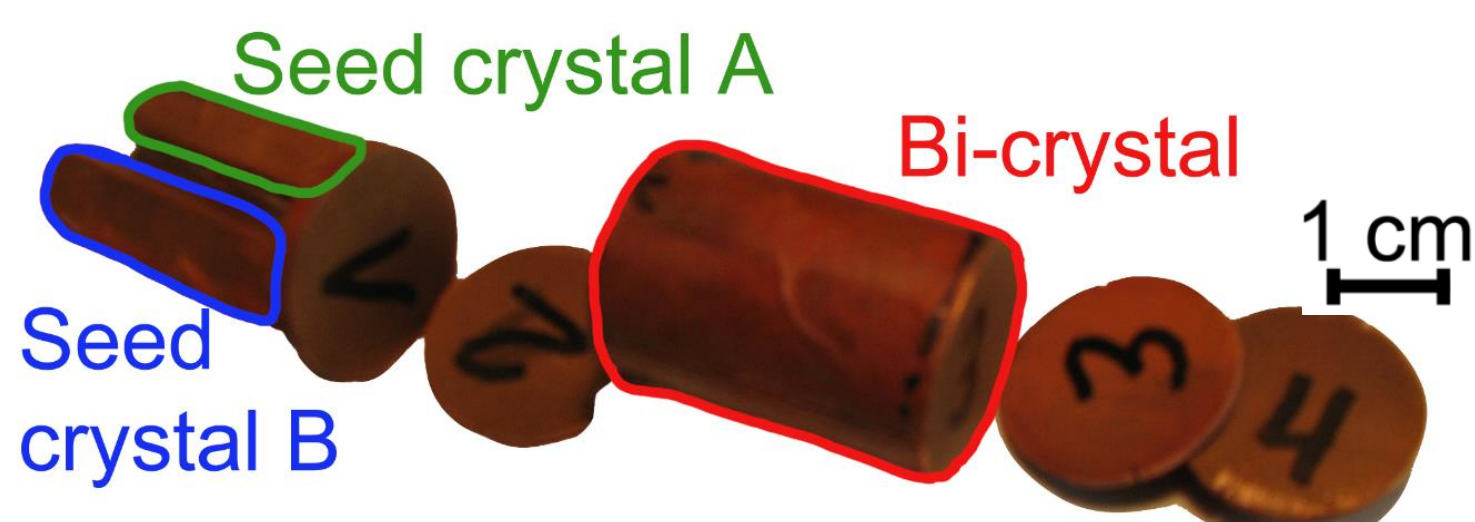
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## Motivation

Dislocation-twin boundary interaction is not entirely understood but gains attention due to the outstanding mechanical performance of nano-twinned materials. Here, we show  $\mu$ Laue compression experiments on a coherent  $\Sigma 3$   $\langle 111 \rangle$  twin. The samples are all tested in different crystallographic loading direction with the twin boundary being parallel to the loading direction.

## Sample production

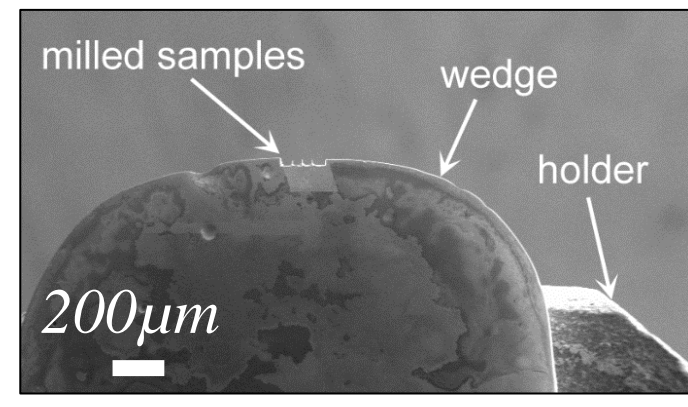
### TB production in Bridgman furnace



- material: **copper**
- using graphite crucible
- growth rate 10 mm/hour

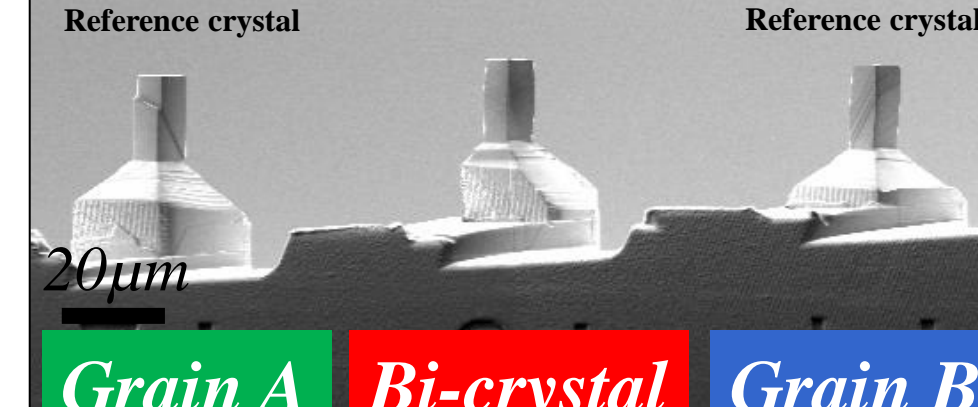
### Meso- & micro sample preparation

#### electrochemical etching



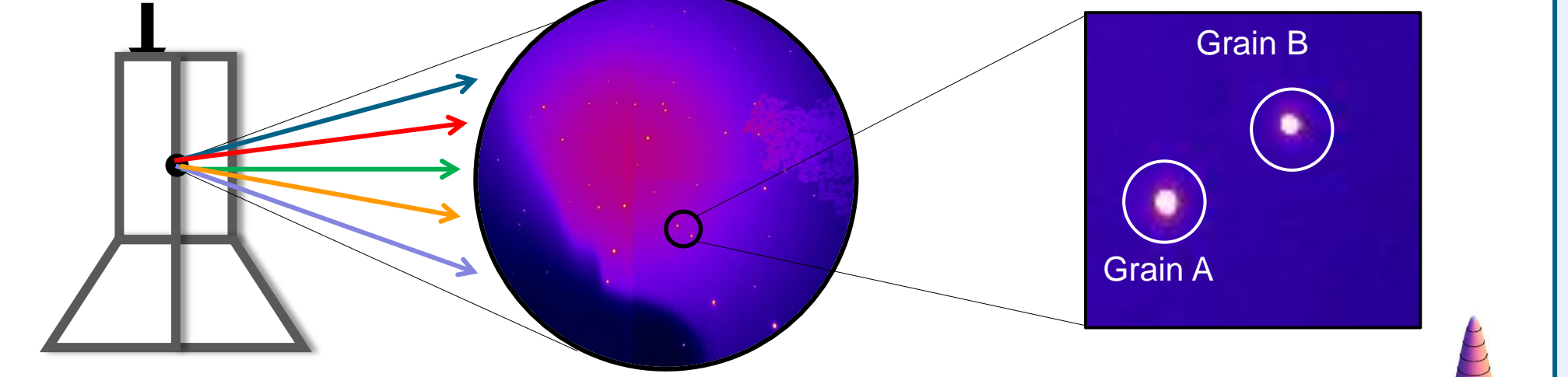
- etching: coarse 15V  
fine 2V
- radius at the tip  $\sim$  few  $\mu$ m

#### FIB milling



- milling: coarse 16 nA  
fine 600 pA
- 5  $\mu$ m
- no taper formation

## in situ $\mu$ Laue compression



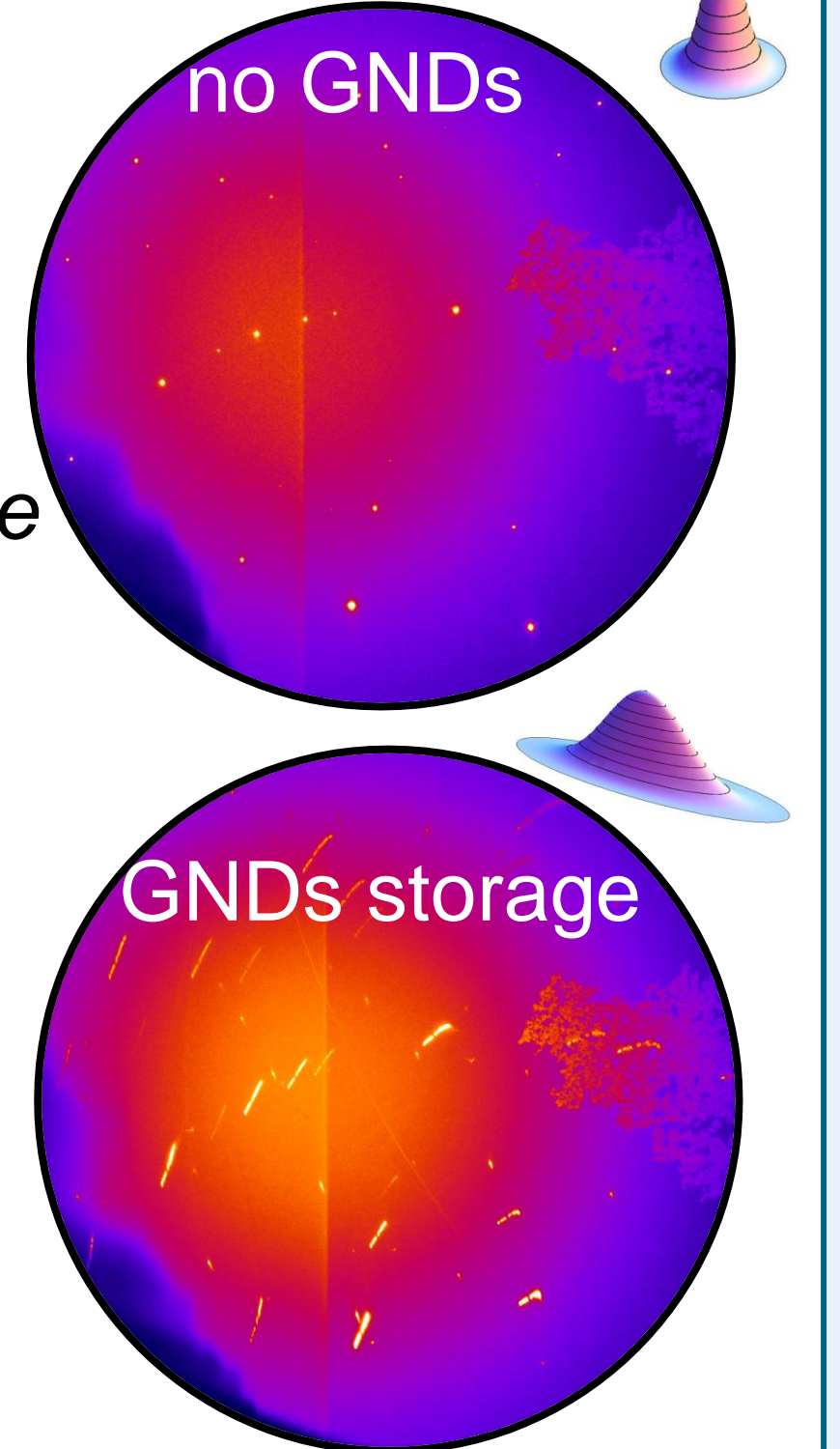
- displacement control mode
- strain rate  $10^{-3} \frac{1}{s}$

Engineering stress strain curve

- crystallographic orientation
- Point to origin misorientation

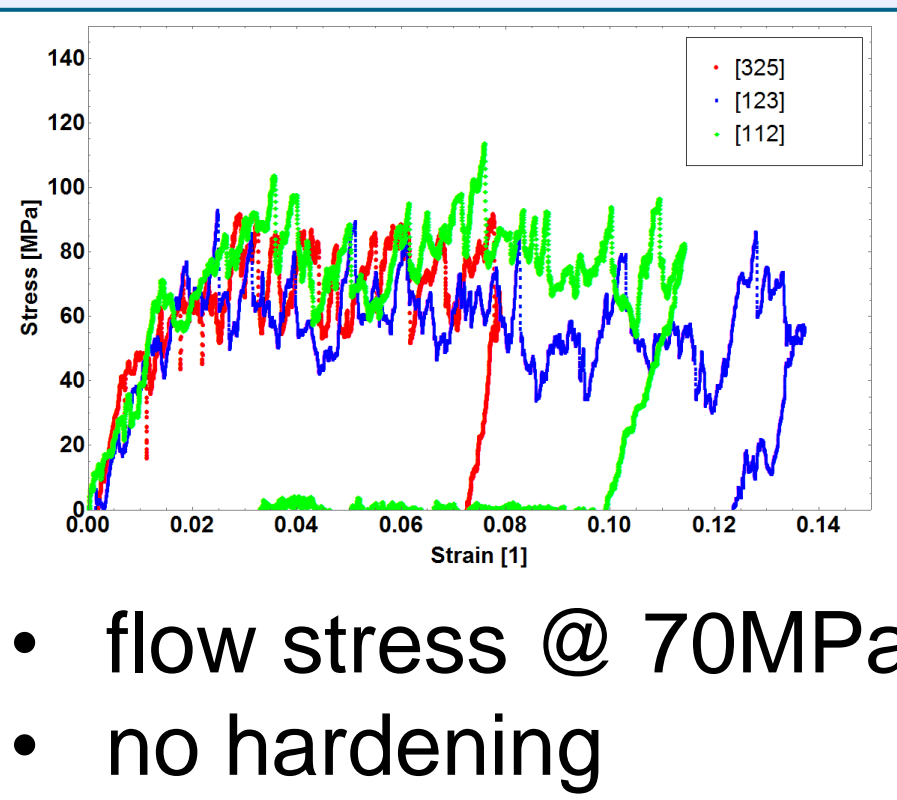
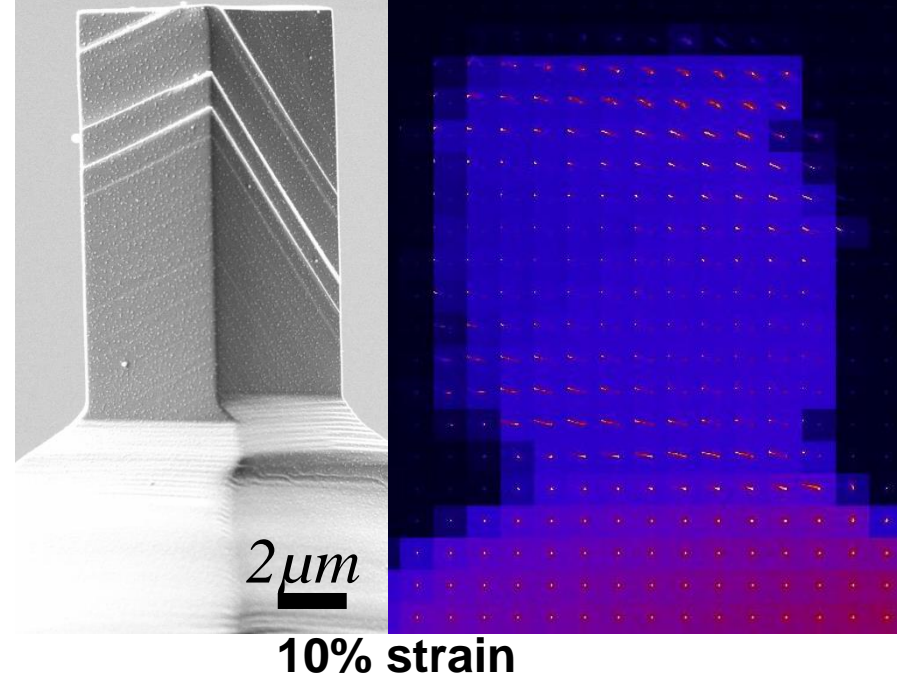
- Peak width
- Estimate the GNDs density

- Streaking analysis (pending)



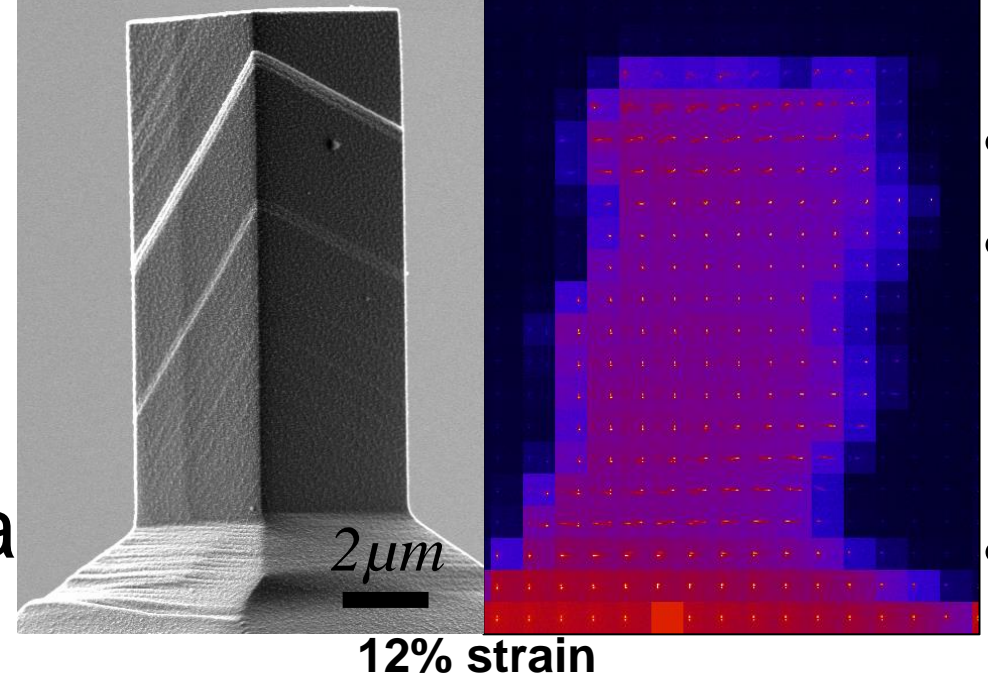
## Single crystals

### grain A



- flow stress @ 70MPa
- no hardening

### grain B



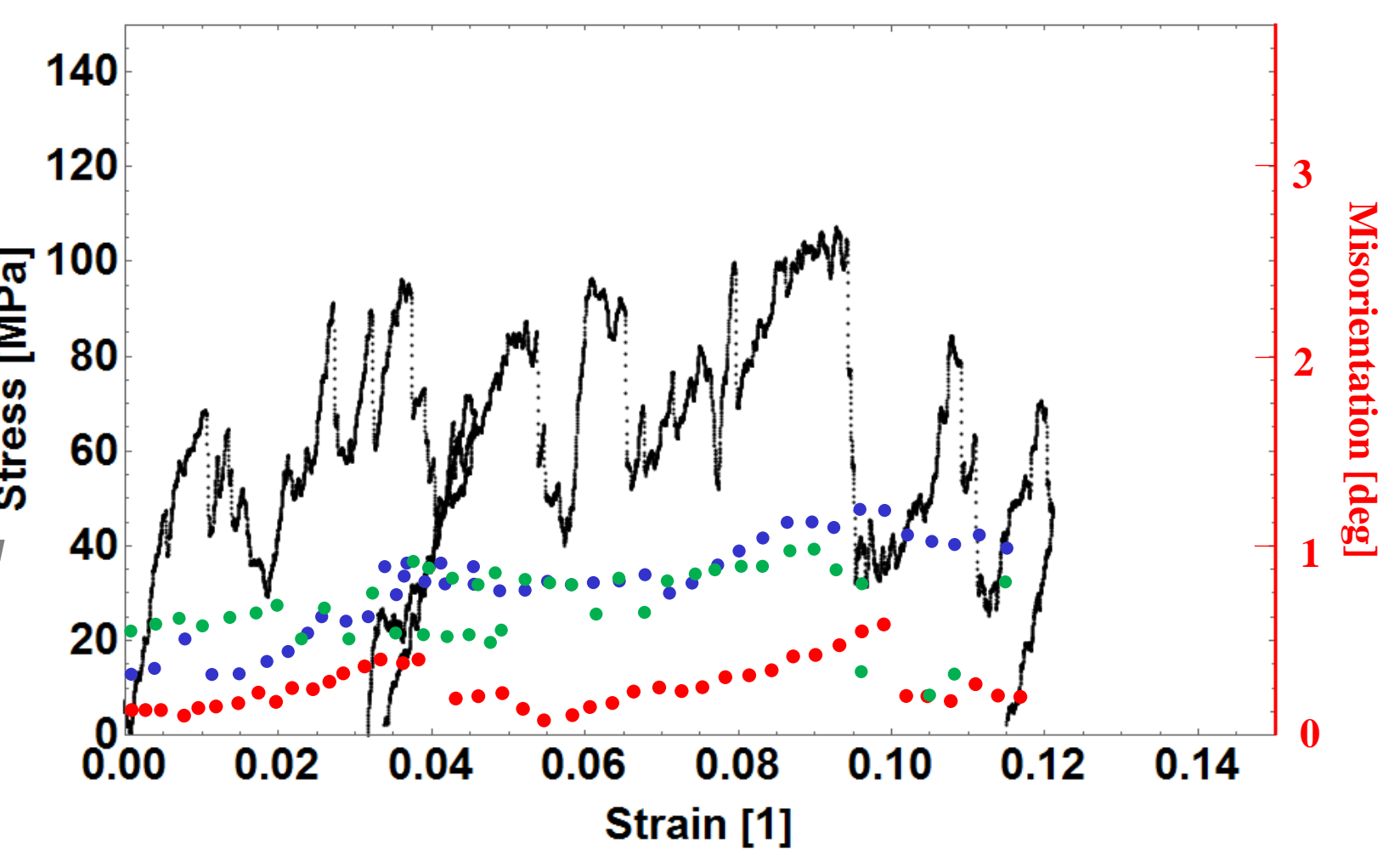
- primary slip system activated (single slip)
- large slip steps formed
- streking only at top and bottom in the same direction  $\rightarrow$  due to instrumental constrains [1]
- no streaking in the center  $\rightarrow$  **low amount of GNDs**

## Samples containing a CTB

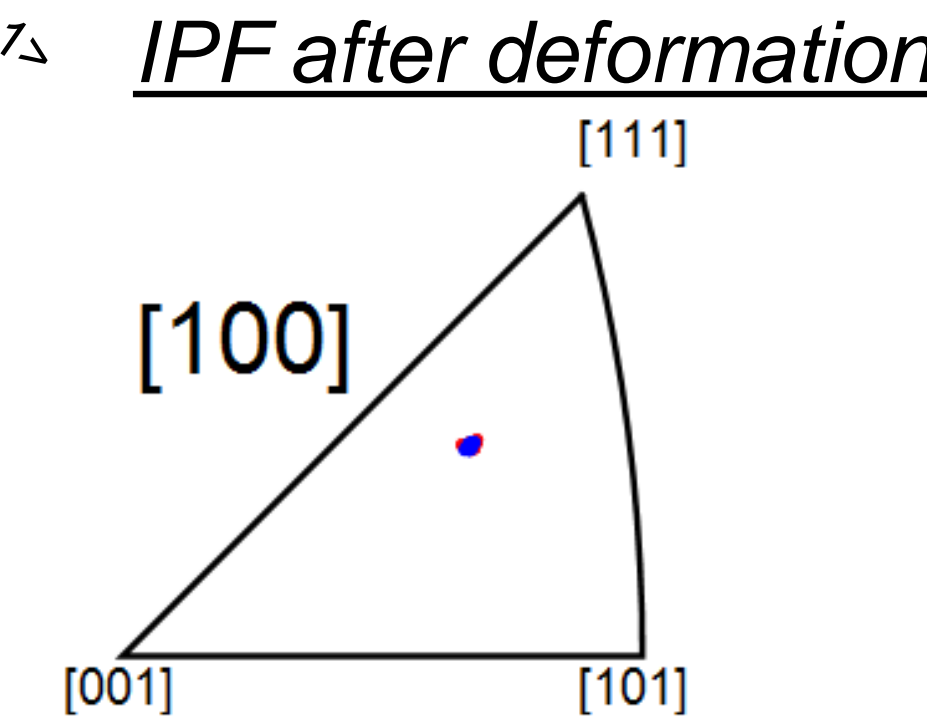
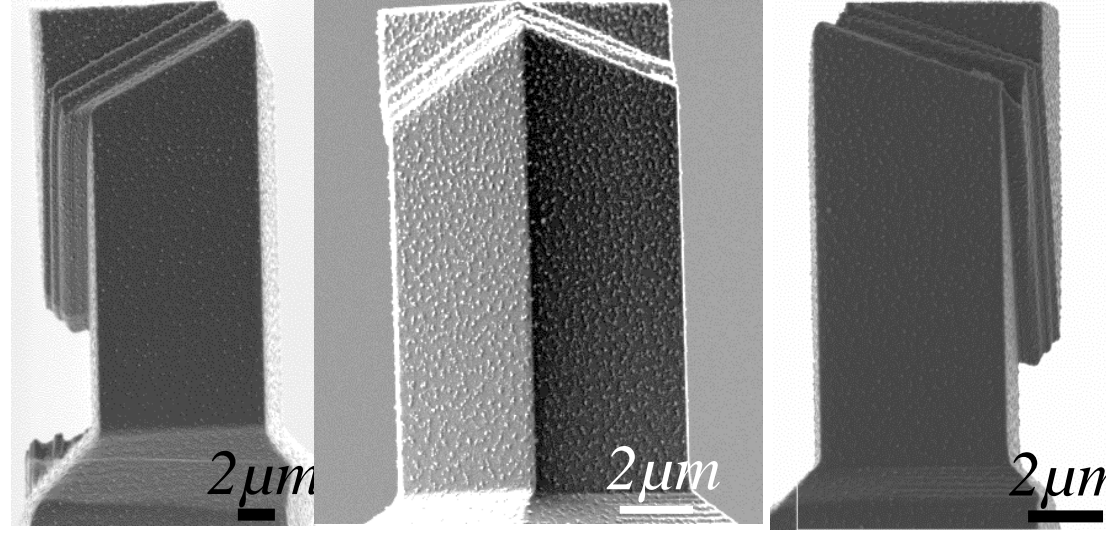
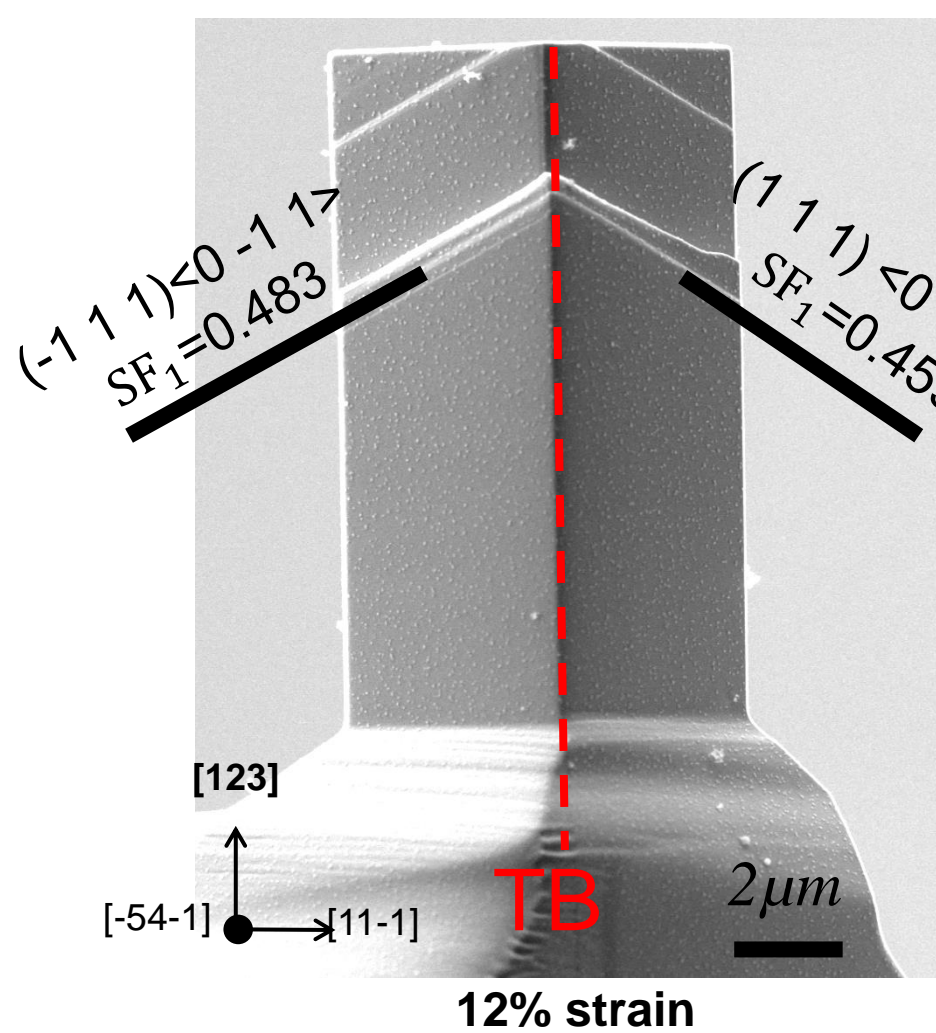
### Stress-Strain Curves

1.

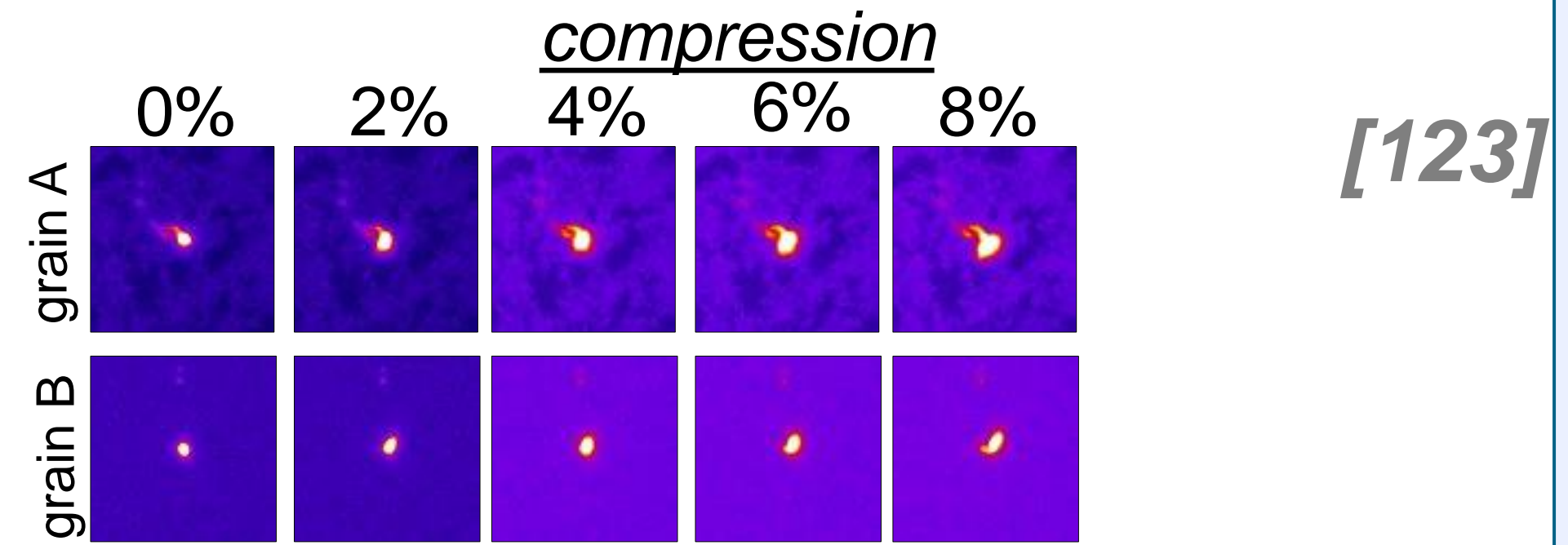
[123]



### SEM after deformation

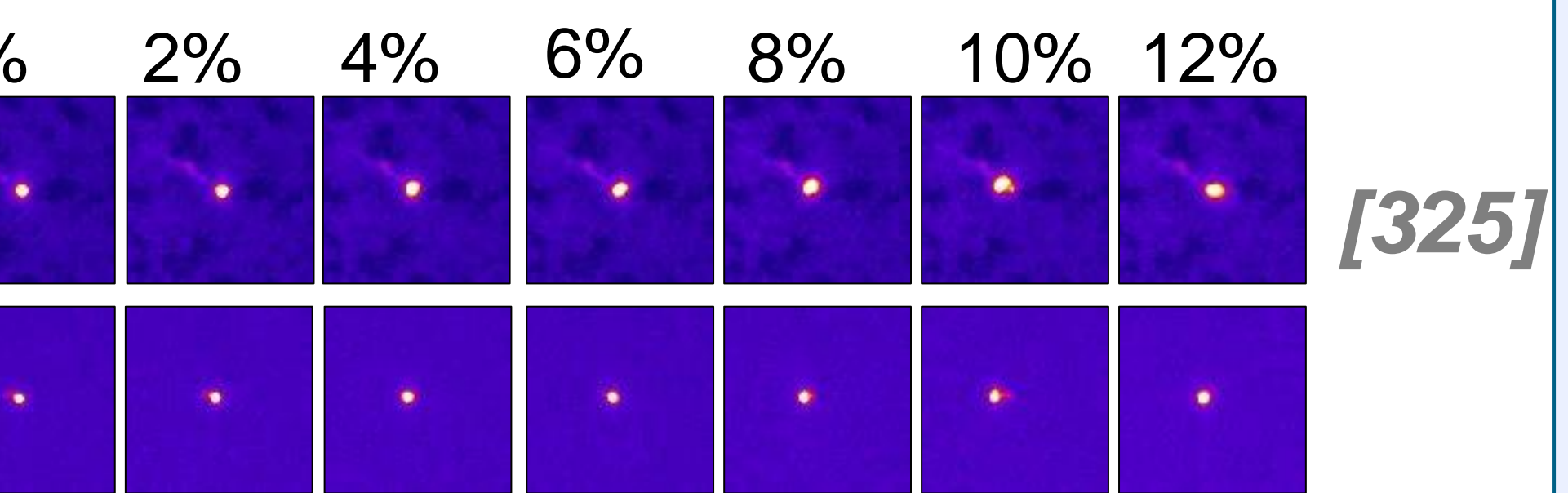
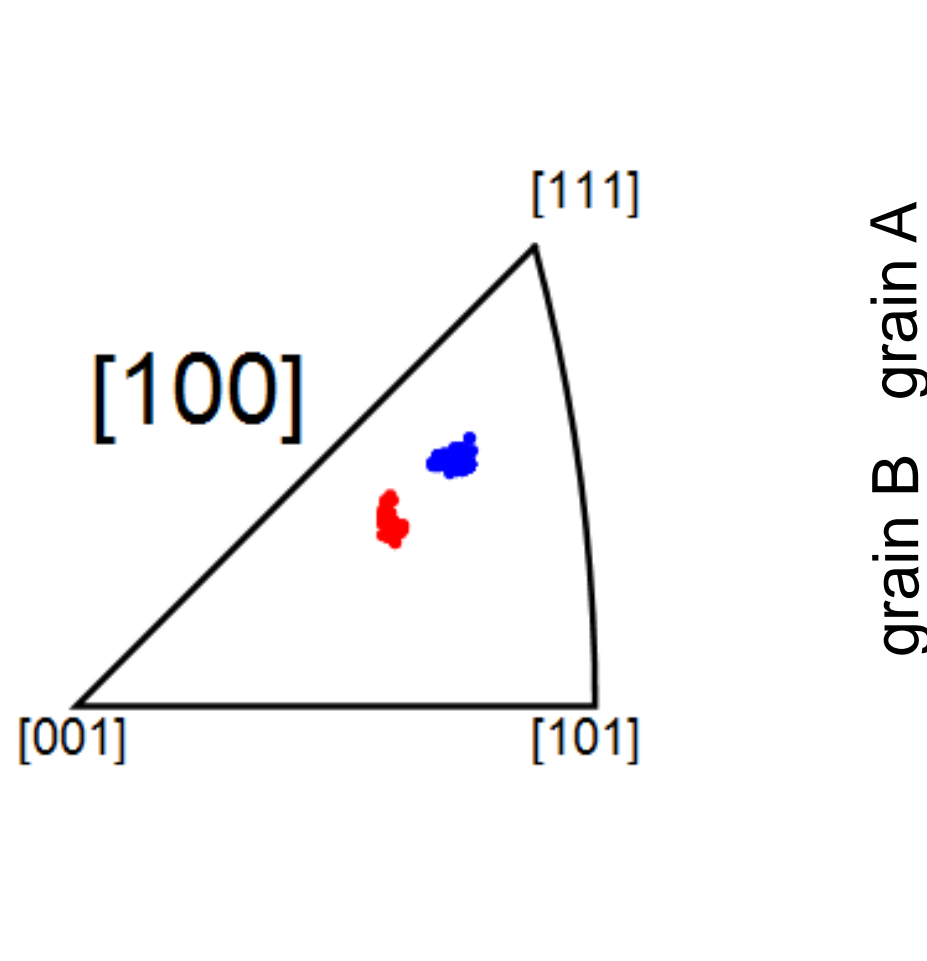
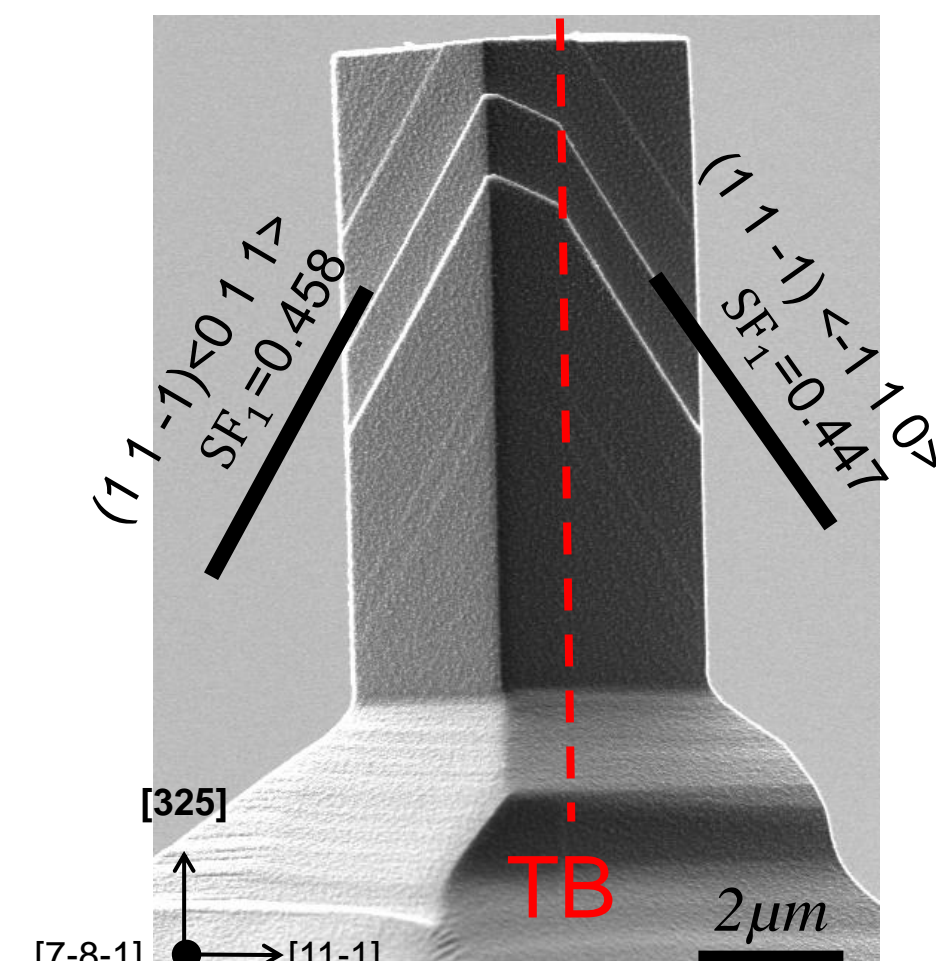
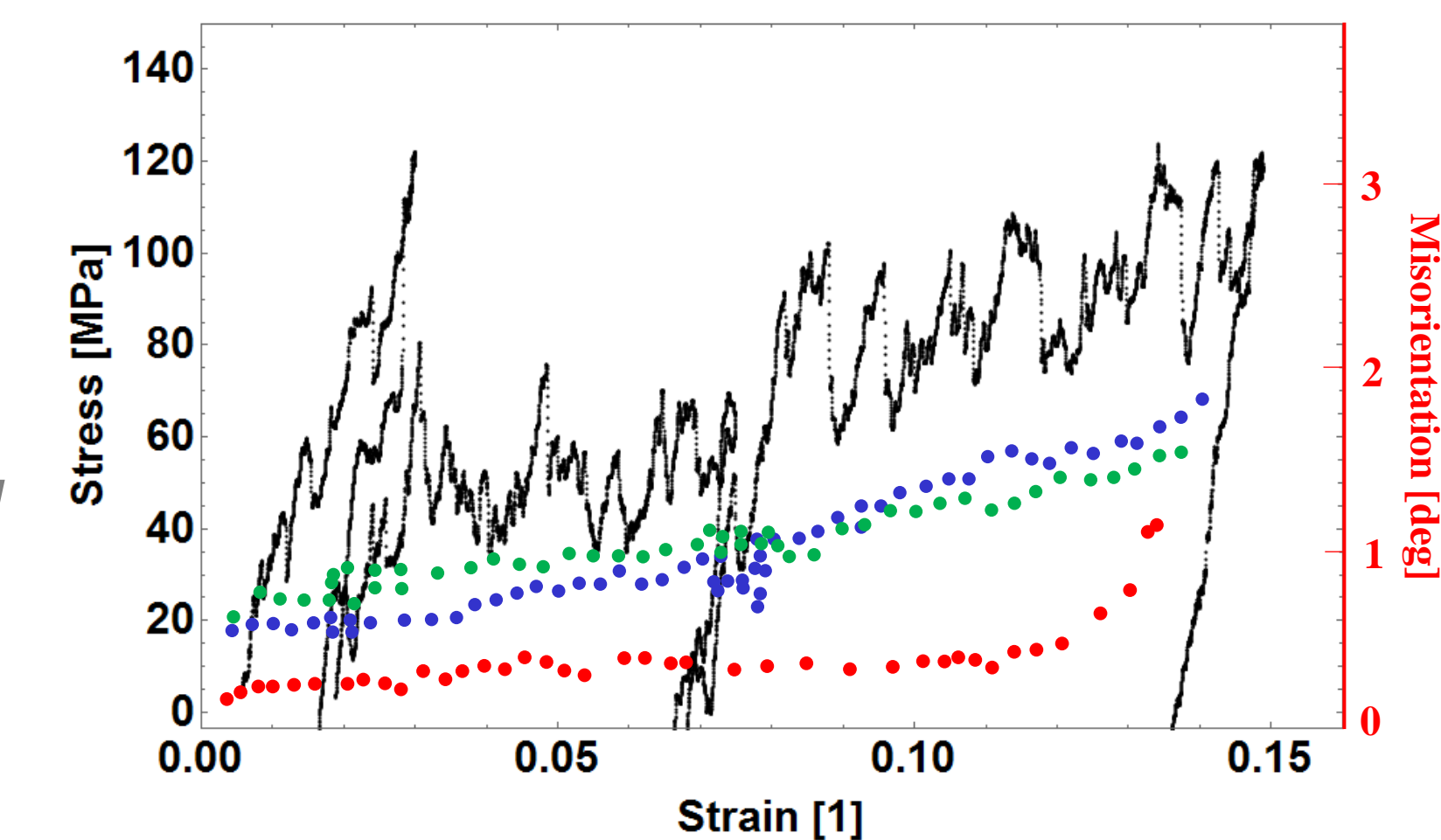


### Laue spot evolution of CTB crystals during compression



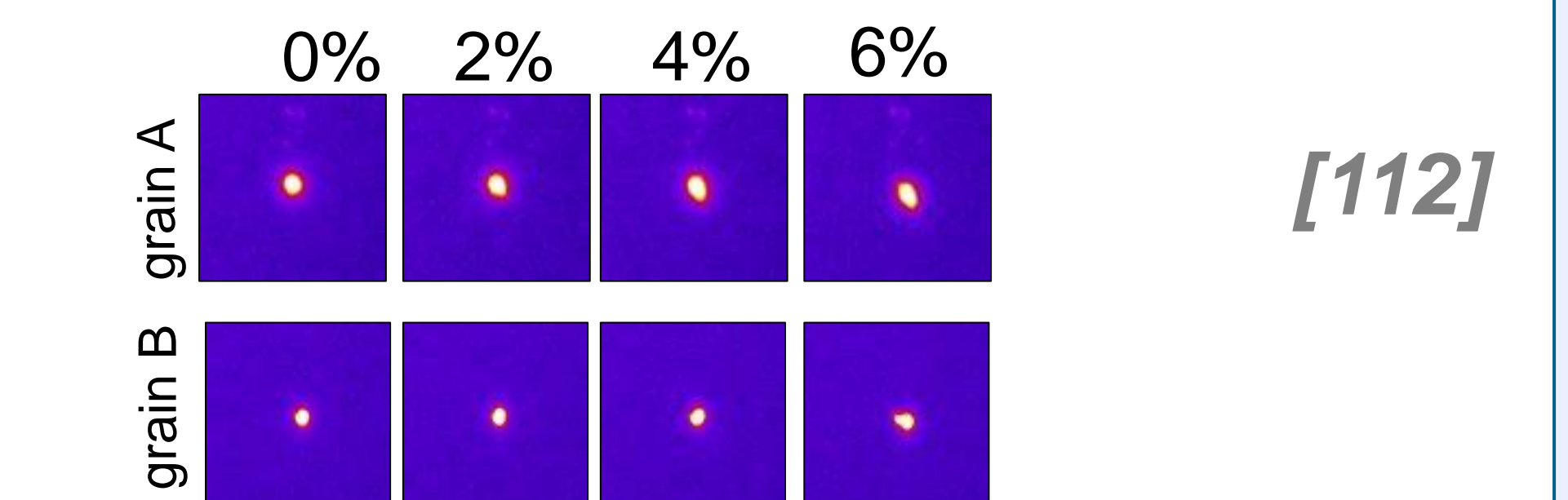
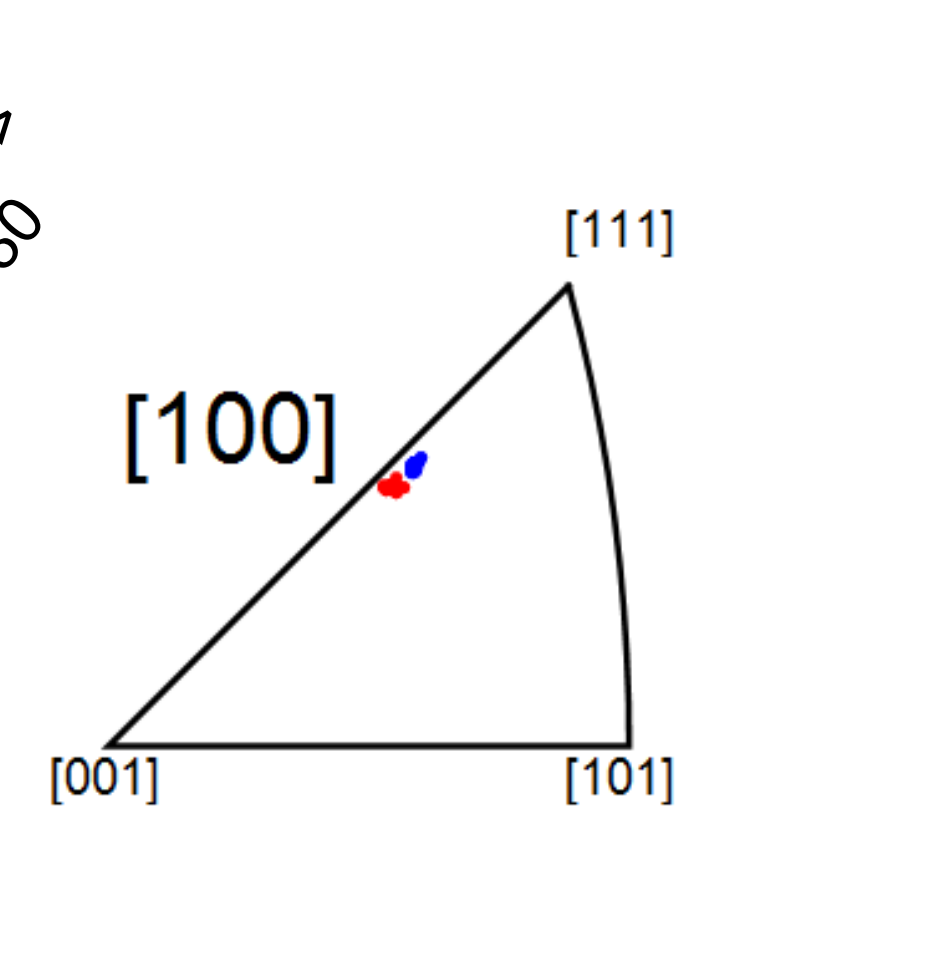
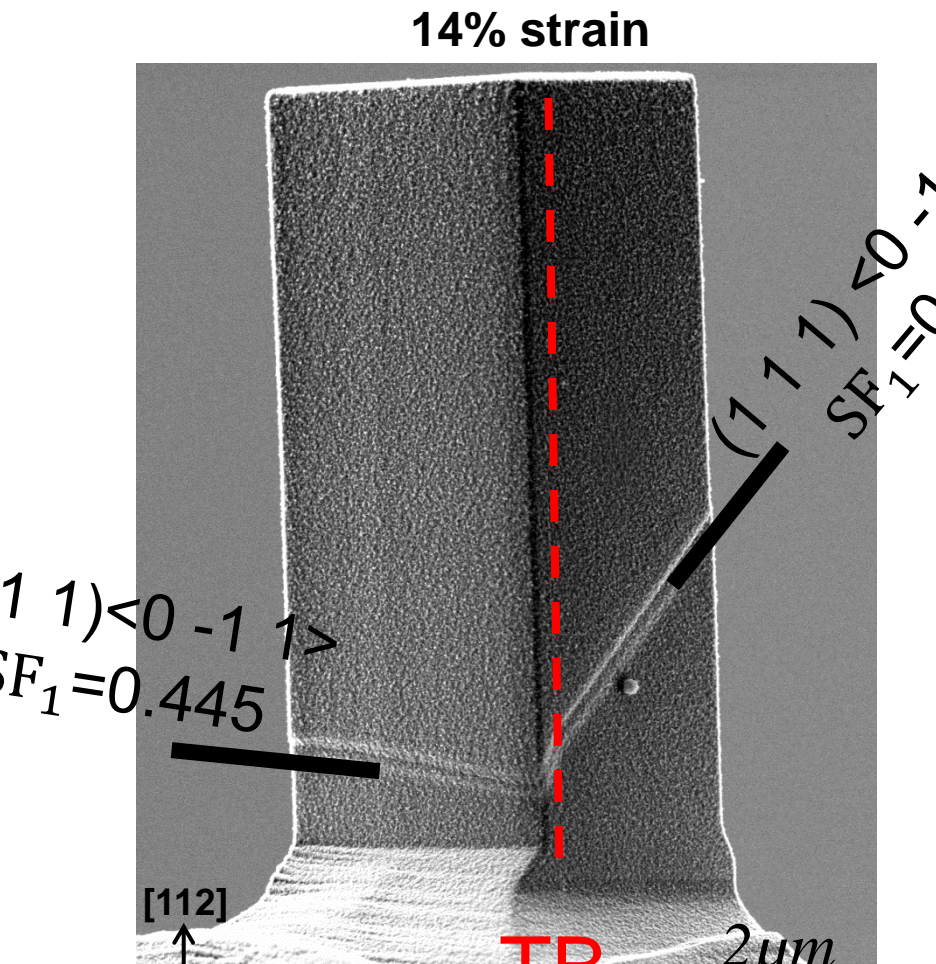
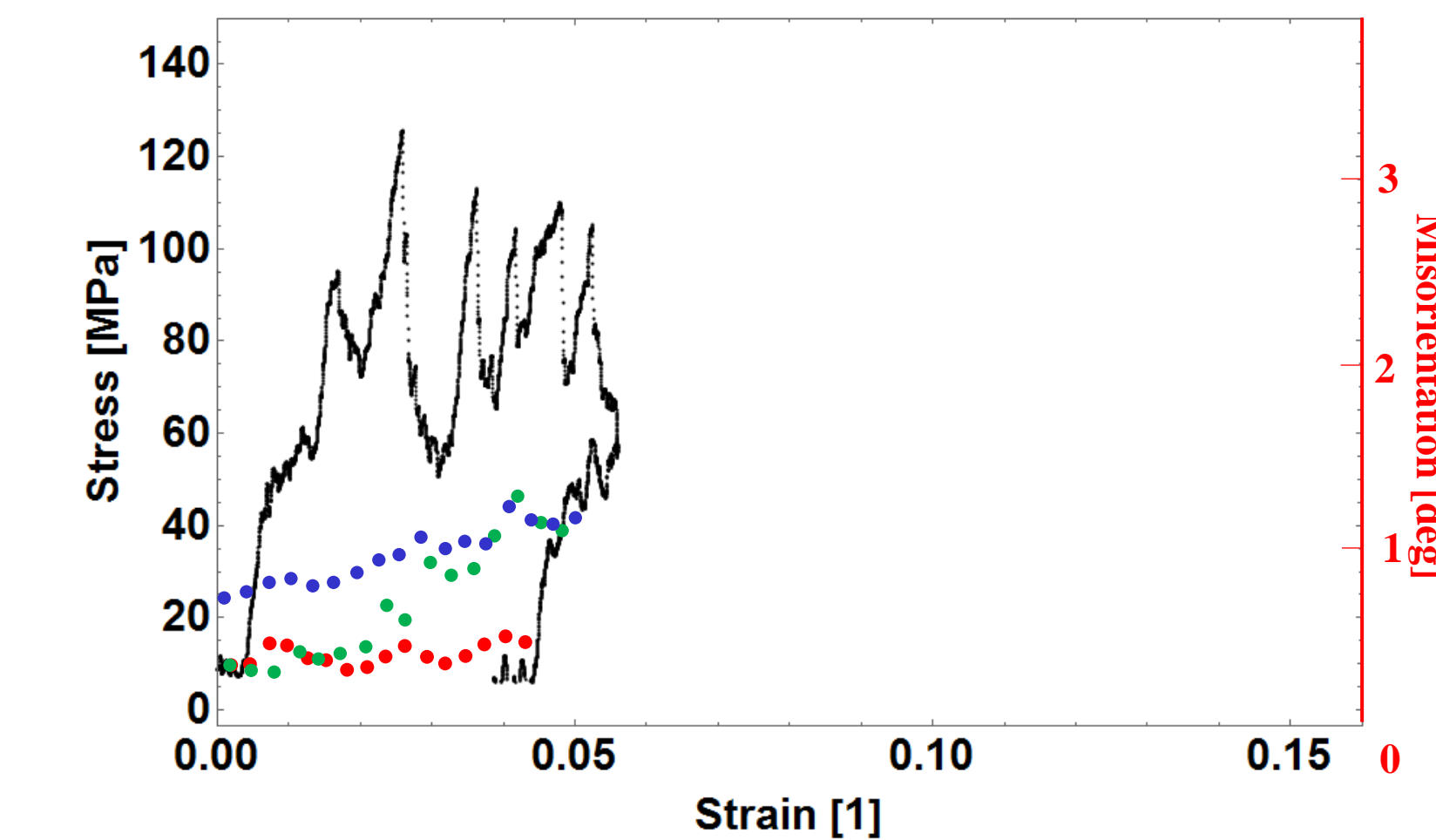
2.

[325]



3.

[112]



- flow stress comparable to  $S_{xx}$
- no hardening observed
- only small change in misorientation ( $\leq 0.5$  grad) below 10% strain
- peak shape stays circular  $\rightarrow$  **low amount of GNDs**

- primary slip system activated (single slip)
- large slip steps
- slip steps meet at TB as observed by Imrich [2]

- peak shape stays unaffected during straining up to about 10% independently of the compression direction
- unresolvable low amount of GNDs**
- "single crystal" like behavior**

## Conclusions

- Stress-Strain behavior, occurrence of the large slip steps and diffraction peak shape during deformation show "single crystal" like behavior
- For all orientations the CTB does not occur as an obstacle for dislocation movement

## References

- [1] Kirchlechner at all Acta Materialia 59, 2011  
[2] Imrich at all, Acta Materialia 73, 2014

## Acknowledgements

- Srividya Subramanian  
Jürgen Wichert  
Gerhard Bialkowski  
MPIE, Düsseldorf, Germany  
Peak shape analysis  
bi-crystalline samples production  
bi-crystalline samples preparation